THE LOW TEMPERATURE SYNTHESIS OF METAL OXIDES BY NOVEL HYDRAZINE METHOD

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The hydroxide, oxalate and citrate precursors of the metal oxides such as γ -Fe₂O₃, (MnZn)Fe₂O₄, Cu(K)Fe₂O₄, BaTiO₃, La(Sr)AnO₃, La(Sr)AlO₃, La/Gd(Ca/Ba/Sr)CoO₃, and anatase TiO₂ on modifications with the hydrazine decompose at low temperatures give single phase oxides of superior properties, while the complexes without such modification require higher temperatures for achieving the phases. The hydrazine released at lower temperatures reacts with the oxygen in the atmosphere, N₂H₄+O₂ \rightarrow N₂+2H₂O; Δ H= -625 kJ mol⁻¹, and liberates enormous energy that is sufficient for the oxidative decomposition of the complexes now devoid of hydrazine. Such extra energy is not available in the case of the precursors without such modifications. The reaction products of hydrazine oxidation provide desired partial pressure of moisture needed for the stabilization of γ -Fe₂O₃. Also, the nitrogen that is formed in the reaction of hydrazine with oxygen gets trapped in the lattice of TiO₂ giving yellow color nitrogen doped TiO_{2-x}N_x photocatalyst. Thus, hydrazine method of preparation has many advantages in the preparation of metal oxides of superior properties.

Keywords: hydrazinate precursors, hydrazine, N-doped oxide, thermal decomposition

Introduction

The recent arousal of interest in nanosize oxide materials is a challenge to the synthetic chemists as it is a material preparation that matters a lot, that too, in the reproducible manner. The development of nanostructure materials with unique parameters is impossible without mastering techniques for producing such weakly agglomerated powders with mean size of 10-15 nm. However, the chemists have always been active in synthesis of submicron size materials of high surface area active oxides that require say, for catalytic applications or for further sintering them into well dense electric, magnetic and optical materials. But, because of their coarse nature these submicron size particles suffer in giving properties that are grain size controlled, such as electrical conductance. The continued synthetic strategies adopted by the chemists have enabled them to achieve materials of average particle size of 100 nm through techniques such as sol-gel, spray-pyrolysis, pulsed laser ablation, chemical vapor deposition, thermal decomposition etc., whereas gas phase methods have resulted into nanosize particles. Then, these advanced methods of preparation of nanosize materials need metal oxide precursors, such as metalloorganic compounds, which are expensive and processes require stringent conditions. Hence, it is always better to explore further the conventional processes, such as thermal decomposition of easily decomposable precursors of oxides. Lowering of temperature of decomposition has

better chance to obtain the desired oxides in very fine particle size which has been realized by the chemists over the years, but it needs furthering of research in this area to get precursors of very low decomposition temperature. And, we have been considering this aspect of synthesizing oxides of spinel, perovskite, anatase structure in our laboratories [1-9] using novel hydrazine modified precursors. Our objective in this type of research was to prepare electric, magnetic, sensing and photoconducting materials of submicron size of average 100-150 nm from low temperature thermal decomposition of metal and mixed metal carboxylates, hydroxides. However, the modification of these precursors by hydrazination enabled us to further decrease the decomposition temperature resulting into sub-submicron size particles of average 50-75 nm with low agglomeration as compared to those obtained without such modifications. Since these are easily decomposable metal complexes, it is but natural to use the thermal analysis techniques to assess their thermal decomposition paths. Here in the present paper we are compiling our results of thermal analysis of metal oxalates, citrates, hydroxides and their hydrazinate precursors of the oxide systems such as, y-Fe₂O₃, (MnZn)Fe₂O₄, CuFe₂O₄, Cu(K)Fe₂O₄ BaTiO₃, La(Sr)MnO₃, La(Sr)AlO₃, La/Gd(Ca/ $Ba/Sr)CoO_3$, and anatase TiO₂.

Hydrazine method of preparation is a novel way of synthesizing materials of desired structure and properties at lower temperatures. For instance, ferrous

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oxalate dihydrate, FeC₂O₄·2H₂O, decomposes in air to hexagonal corundum nonmagnetic α -Fe₂O₃, while the hydrazinated oxalate, FeC2O4·2N2H4, auto-catalytically decomposes to cubic magnetic γ -Fe₂O₃ [1–2] and similar observations were also made for other iron(II) carboxylates such as ferrous-fumarate/ succinate/maleate/malate/malonate [3]. Iron hydroxides modified by hydrazine too showed the easy formation of γ -Fe₂O₃ [4]. Ni–Zn–Ferrite [5] and MgFe₂O₄ [6] synthesized from oxalate precursors that modified by hydrazine showed better magnetic properties. Dielectric properties of MgFe₂O₄ [7] from oxalate hydrazinate and semiconducting properties of y-Fe₂O₃ from iron hydroxides modified by hydrazination [8] are found to be superior as compared to the oxide prepared from the unhydrazinated compound. The titanium hydroxide and titanium oxalate decompose to white pigment grade anatase TiO₂, while their hydrazinates yield yellow color visible light sensitive nitrogen doped TiO_{2-x}N_x photo-catalyst [9] showing improved electrical properties [10]. In hydrothermal route for synthesizing SnO₂ quantum dots [11] the use of hydrazine hydrate is most critical in achieving the particles in a small range of 2.3–3.1 nm. In room temperature preparation of novel Cu_{2-x}Se nanotubes in organic solvents the presence of hydrazine hydrate aids the formation [12]. A novel low-temperature sol-gel based process in the synthesis of crystalline in the Al-N-O system involves a reaction between aluminum tri-sec-butoxide with hydrazine hydrate [13] in the acetonitrile and chloroform medium. Conventional high temperature method of preparation of CoS has been brought down to low temperature by hydrazine assisted synthetic process [14].

Experimental

Synthesis

Oxalate/citrate precursors

The oxalate precursors of spinels, MFe₂O₄ (M=Cu/Mn–Zn/K); perovskites, La/Gd (M^{II})MO₃ (M=Mn/Co/Al; M^{II} =Ca/Sr/Ba) and BaTiO₃, and TiO₂ were synthesized by standard methods. Commercial metal chlorides and oxalic acid/citric acid were used for the synthesis. However, in the preparation of the oxalate precursor of TiO₂ a freshly prepared titanium chloride from commercial TiO₂ was used. Here the commercial titania was fused with KOH in Ni crucible and then by adding conc. HCl, potassium was removed as KCl and the fresh titanium chloride that now obtained was treated with ammonia to get titanium hydroxide. The titanium chloride.

Hydroxide precursors

Titanium hydroxide was prepared by taking titanium iso-propoxide in iso-propanol and adding slowly to a large quantity of water. Titanium hydroxide was also prepared by adding ammonium hydroxide to titanium chloride.

Modification of oxalate/citrate/hydroxide precursors by hydrazine

a) Under hydrazine (UH) method: equilibration: The oxalate and hydroxide precipitates were spread over a Petri dish and placed in closed vessel containing 99.9% hydrazine hydrate. The hydrazine intake was monitored by titrimetrically using KIO₃ titrant [15].

b) Synthetic hydrazine (SH) method: In another method of preparation of hydrazine modified precursors, the oxalic acid and hydrazine hydrate were first stirred well under inert nitrogen atmosphere in 3 necked retort and to the mixture a freshly prepared metal chloride solution was rundown to precipitate out the hydrazinated complexes.

Methods

Chemical, isothermal mass loss analysis: formula fixation

For proposing chemical formula for the oxalate, hydroxide, citrates and their hydrazinate complexes chemical estimation of metals, oxalate and hydrazine were done as per the standard methods described in Vogel [15]. Total mass loss measurements were done in a muffle furnace. Infrared analyses were done on Shimadzu IR Prestige-21.

Thermal analysis

Thermogravimetric analysis (TG) and differential scanning calorimetric (DSC) analysis were preformed on NETZSCH DSC-DTA-TG STA 409PC. The heating rate was fixed at 10°C min⁻¹. The analyses were done in the flowing air atmosphere in most of the cases, and in few cases nitrogen gas was used.

X-ray diffraction (XRD), SEM/TEM and XPS studies

The decomposed products were identified from X-ray diffraction (XRD) studies on ITAL APD 2000 using CuK_{α} radiation (λ =1.5418 Å) and Ni filter. The d_{hkl} values obtained by XRD were matched with the JCPDS/ICDD Card of individual oxide systems. Scanning electron microscopic (SEM) studies were conducted on HITACHI S-4500. Transmission electron micrographs (TEM) were recorded with JEOL-JEM 100SX microscope, working at 100 kV

accelerating voltage. X-ray photoelectron spectroscopy (XPS) measurements were carried out over a Kratas Axis Spectrometer at a vacuum of $3 \cdot 10^{-9}$ Torr with non-monochromatic MgK_a radiation.

Nitrogen estimation, dielectric property, saturation magnetization, electronic conductivity

The amount of nitrogen in TiO_2 was determined using the oxygen-nitrogen analyzer [16] (HIROBA, EMGA, 2800). Dielectric constant of BaTiO₃ was measured as a function of temperature in a two-probe set up in a constant frequency of 1 KHz and that of LaAlO₃ at different temperatures and frequencies. Saturation magnetization of magnetic CuFe₂O₄ in emu/g was carried out on the instrument supplied by M/s. Arun Electronics, Mumbai, [17] using Ni as standard. Electronic conductivity of LaCoO₃ system was measured by Van der Pauw method [18]

Results and discussion

Chemical formulas

The observed percentage of oxalate, citrate, hydrazine and total mass loss of the complexes are compiled in the Table 1. From these values formula for each complex is proposed and the calculated values given in the bracket are found to match well with observed ones. The infrared (IR) spectra of all these complexes were recorded in the range of 4000–400 cm⁻¹ and important peaks have been listed in the Table 1. The carboxylate complexes show absorption frequencies in the range 3000–3600 cm⁻¹ which may be attributed to O–H frequency (v_{OH}) of water, but more broadening of the peaks observed for the hydrazinated complexes which is due to the N-H stretching (v_{NH}) frequencies. The carboxylate groups have their asymmetric and symmetric stretching frequencies, v_{ooc} asym/sym, in the region 1640–1610 and 1450–1375 cm⁻¹, respectively. The observed N–N stretching (v_{NN}) frequencies in the region \sim 980 cm⁻¹ indicate the presence of hydrazine, N₂H₄, either as $N_2H_5^+$ or as a bidentate bridging H_2N-NH_2 [19–21]. The NH₂ bending vibrations at ~1600 cm⁻¹ (which could include contributions from δ_{NH}) and 1540 cm⁻¹ [22-24] although merged in the hydrazinate complexes of the carboxylates, band positions in this range clearly indicate its presence in the titanium hydroxides. The fundamental stretching of hydroxyl (free or bonded) frequencies in the range 3600-3100 cm⁻¹ in the titanium hydroxide are observed in combination with another peak $\sim 1630 \text{ cm}^{-1}$ due to the bending vibrations of coordinated H₂O as well as Ti-OH [26]. However, a peak ~1400 cm⁻¹ that observed has been assigned to

[26] the oxyhydroxide. A broad stretching vibration mode of OH group at 3425 cm^{-1} and corresponding bending vibration band at 1637 cm⁻¹ observed within a TiO₂ sol-gel [27] occurred during the gelling of the titanium alkoxide. An additional band at 1389 cm⁻¹ that had been assigned to an asymmetric bending vibration of C-H [27] due to the presence of organic moiety is not found in our titanium hydroxide prepared by the hydrolysis of titanium isopropoxide, Fig. 1a, but a broad band centered \sim 3340 cm⁻¹ due to stretching vibration and corresponding bending vibration band at 1641 cm⁻¹ have appeared. On the other hand, the hydroxide prepared from freshly prepared titanium chloride by precipitating with ammonia, shows, Fig. 1b, an additional band at 1427 cm⁻¹ along with very broad band centered ~3240 and 1631 cm⁻¹. However, infrared spectra recorded on the hydroxides prepared from the titanium chloride and heat treated at different temperatures up to 800°C, Figs 1a and b, indicate that the band at 1427 cm⁻¹ vanishes and the broad band centered ~3240 cm⁻¹ narrows at 300°C and finally at 800° C the ~3240 cm⁻¹ band shows its weak presence.



Fig. 1 Temperature variation of FTIR of titanium hydroxide obtained from a – titanium isopropoxide, b – titanium chloride

Proposed chemical formula	Total mass loss/%	C2O4/ %	N2H4/ %	v_{OH}/v_{NH}	IR peak/cm ⁻¹ v_{ooc} / asym/sym δ_{NH}	$v_{ m NN}/\delta_{ m NN}$
$LaAl(C_2O_4)_3 \cdot 10H_2O$	64.5	43.0	_	3200-3400	1650/1325	_
	(64.9)	(43.3)				
LaAl(C ₂ O ₄) ₃ ·N ₂ H ₄ ·3H ₂ O	64.9	51.3	6.7	3200-3500	1625/1325	960
	(64.7)	(51.2)	(6.2)			
La _{0.65} Sr _{0.35} Al(C ₂ O ₄) ₃ ·2N ₂ H ₄	68.2	44.0	11.0	3200-3600	1625/1350	960
	(68.2)	(43.9)	(10.6)			
$La_{0.7}Sr_{0.3}Mn(C_2O_4)_3 \cdot 2N_2H_4 \cdot 15H_2O$	70.9	34.8	8.2	3200-3550	1625/1350	960
	(70.8)	(34.0)	(8.2)			
$GdCo(C_6H_5O_7)_2 \cdot 2H_2O$	75.8	_	_	3400	1600/1400	_
	(75.7)					
$GdCo(C_6H_5O_7)_2 \cdot 4N_2H_4$	74.0	_	17.6	3300-3500	1700/1400	960
	(73.2)		(17.7)			
$Gd_{0.8}Ca_{0.2}Co(C_6H_5O_7)_2 \cdot 10H_2O$	74.0	_	_	3300-3500	1700/1400	
	(74.3)					
$Gd_{0.8}Ca_{0.2}Co(C_6H_5O_7)_2 \cdot 4N_2H_4$	72.5	_	17.9	3300-3500	1750/1400	980
	(72.8)		(18.1)			
Gd _{0.8} Ba _{0.2} Co(C ₆ H ₅ O ₇) ₂ ·10H ₂ O	78.4	_	_	3300-3500	1750/1400	_
	(79.1)					
$Gd_{0.8}Ba_{0.2}Co(C_6H_5O_7)_2 \cdot 4N_2H_4$	76.0		13.6	3300-3500	1750/1400	980
	(75.2)		(13.9)			
$Ba[TiO(C_2O_4)_2] \cdot 6H_2O$	52.0	59.0	_	3600-3400	1680/1420	
	(51.9)	(59.8)	_			
$Ba[TiO(C_2O_4)_2]$ ·7N ₂ H ₄	62.7	29.1	37.0	3500-3300	1680/1425	1084
	(61.2)	(29.2)	(37.2)			
$(NH_4)_2TiO(C_2O_4)_2\cdot H_2O$	72.2	59.2	-	2800-3500	1600/1300	_
	(72.8)	(59.8)				
(N ₂ H ₅) ₂ [TiO(C ₂ O ₄) ₂]·N ₂ H ₄ (UH)	76.8	52.1	9.8	2200-3200	1600/1300	980
	(76.3)	(52.1)	(9.4)	Very broad		
(N ₂ H ₅) ₂ [TiO(C ₂ O ₄) ₂]·3N ₂ H ₄ (SH)	79.4	43.2	23.5	2600-3200	1700/1300	980
	(79.9)	(43.8)	(23.8)	Very broad	1500	
TiO(OH) ₂ ·2.5H ₂ O	44.1	_	_	3200-3600	1400	_
	(45.0)	_	_		1630	
TiO(OH) ₂ ·N2H ₄ (UH)	40.0	_	24.1	3200-4000	1400	_
	(39.0)	_	(24.6)	_	1630	
TiO(OH) ₂ ·4N ₂ H ₄ (SH)	66.0	_	56.5	3200-4000	1400	_
	(65.0)	_	(56.6)	_	1630	
$CuFe_2(C_2O_4)_3 \cdot 8H_2O$	59.4	45.9	_	3500-3200	1650/1450	_
	(59.0)	(45.2)				
$CuFe_2(C_2O_4)_3 \cdot 3N_2H_4 \cdot 4H_2O$	59.9	46.0	16.4	3500-3200	1650/1450	980
	(59.8)	(44.3)	(16.1)			
$Cu_{0.9}K_{0.1}Fe_2(C_2O_4)_3 \cdot 9H_2O$	61.6	43.7	_	3500-3200	1650/1450	_
0,, 0,1 <u>2</u> <u>2</u> 7/J <u>2</u>	(60.2)	(43.3)				
$Cu_{0.9}K_{0.1}Fe_2(C_2O_4)_3\cdot 5N_2H_4\cdot 9H_2O$	64.5	43.7	24.8	3500-3200	1650/1350	980
	(64.4)	(39.3)	(23.9)			

Proposed chemical formula	Total mass loss/%	C2O4/ %	N2H4/ %	$ u_{OH}/ u_{NH}$	$ \begin{array}{l} IR \ peak/cm^{-l} \\ \nu_{ooc} \\ asym/sym \ \delta_{NH} \end{array} $	v_{NN}/δ_{NN}
$Mn_{0.65}Zn_{0.35}Fe_2(C_2O_4)_3{\cdot}8H_2O$	59.4	46.0	_	3500-3200	1650/1450	_
	(60.2)	(45.8)				
$Mn_{0.65}Zn_{0.35}Fe_2(C_2O_4)_3\cdot 3N_2H_4\cdot 5H_2O$	61.5	28.0	15.3	3500-3200	1650/1350	980
	(62.2)	(28.3)	(15.5)			
$FeC_2O_4 \cdot 2H_2O$	55.8	49.1	-	3500	1650/1350	_
	(55.6)	(48.9)				
FeC_2O_4 ·2N ₂ H ₄	61.8	43.1	31.0	3350	1650–1350	980
	(61.5)	(42.3)	(30.8)			

Table 1 Continued

The numbers in parantheses mean calculated data

But the band at 1631 cm^{-1} also persists at 800°C , while in the case of the hydroxide from alkoxide the bands ~ 3340 and 1641 cm^{-1} vanish at 500°C . These observations suggest that the additional peak at 1427 cm^{-1} and a broad band centered $\sim 3240 \text{ cm}^{-1}$ in the titanium hydroxide prepared by precipitating titanium chloride with ammonia may be due to the presence of ammonia in the precipitate.

On the basis of the results of IR and chemical analysis (presented in Table 1) it appears that hydrazine complexes are formed. The chemical formulas for the GdCo complexes were difficult to ascertain, and an adduct type [25] hydrazine complex of the citrates may be considered in the following manner: $GdCo(C_6H_5O_7)_2 \cdot 4N_2H_4$, $Gd_{0.8}Ca_{0.2}Co(C_6H_5O_7)_3 \cdot 4N_2H_4$, $Gd_{0.8}Ba_{0.2}Co(C_6H_5O_7)_3 \cdot 3N_2H_4$.

Thermal analysis

TG-DSC traces of individual samples have been analyzed separately to record the different peak temperatures and corresponding temperature ranges of mass losses. However, for the purpose of the comparison of the thermal paths of hydrazinate and unhydrazinate precursors of each system, the DSC/TG traces have been shown in one figure. Here, note that the numerical for both the temperature and DSC (mW/mg)/% mass loss axes are arbitrary, not to the scale. The peak temperatures shown are from the individual traces

Spinel systems

CuFe₂O₄ system

The TG-DSC traces of $CuFe_2(C_2O_4)_3 \cdot 8H_2O$ ($CuFe_oxalate$) and $CuFe_2(C_2O_4)_3 \cdot 3N_2H_4 \cdot 4H_2O$ ($CuFe_oxalate_hyd$) are shown in Fig. 2a. The oxalate complex indicates a sharp endothermic peak at ~204°C which is immediately followed by a broad intense exothermic reaction in the range 204–332°C with a peak at ~ 307° C, while its hydrazinate complex shows an exothermic peak at ~ 146° C and another exotherm between 185 and 330°C with a peak at ~ 270° C. The TG mass loss curves for both the complexes between room temperature (rT) and 330°C show two steps decomposition, but, the hydrazine complex clearly shows measurable two-step mass loss. The first step in the oxalate complex is dehydration which is then followed by decarboxylation. The first step on mass loss in hydrazine complex is found to be due to dehydrazination as confirmed by the absence of hydrazine in the complex that had been isothermally



Fig. 2 TG/DSC plots of a – copper iron oxalate/oxalate hydrazinate, b – potassium doped copper iron oxalate/oxalate hydrazinate

decomposed separately at ~130°C. The second step in the hydrazine complex is due to dehydration and decarboxylation. Thus, in the temperature range rT-330°C, the dehydration is followed by decarboxylation in $CuFe_2(C_2O_4)_3 \cdot 8H_2O$, while in the hydrazinate complex, CuFe₂(C₂O₄)₃·3N₂H₄·4H₂O the dehydrazination-dehydration-decarbsequence is oxylation. At ~330°C both the complexes show almost complete decomposition as observed in the TG trace, beyond this temperature there is no measurable mass change, however, DSC shows a broad exothermic up to 1020°C and then there appears an endothermic peak ~1030°C in both samples. The TG trace which was showing a negligible mass loss till 1020°C suddenly indicates a depression in the mass loss and corresponding to that there appears an endotherm $\sim 1030^{\circ}$ C. From the XRD studies, Fig. 3, the thermal product of $CuFe_2(C_2O_4)_3 \cdot 3N_2H_4 \cdot 4H_2O$ that had been isothermally heated separately ~290°C indicated a single phase CuFe₂O₄ formation, while ~330°C is required for $CuFe_2(C_2O_4)_3 \cdot 8H_20$ to form the spinel. Hydrazine modification of the oxalate precursor enables to prepare the spinel at low temperature.

K-doped CuFe₂O₄

The DCS-TG traces of potassium doped oxalate and hydrazinate complexes, $Cu_{0.9}K_{0.1}Fe_2(C_2O_4)_3 \cdot 9H_2O$, $(Cu_{0.9}K_{0.1} \text{ oxalate}), Cu_{0.9}K_{0.1}Fe_2(C_2O_4)_3 \cdot 5N_2H_4 \cdot 9H_2O,$ (Cu_{0.9}K_{0.1}_oxalate_hyd) are shown in Fig. 2b. The oxalate after showing a weak endothermic peak ~199°C immediately indicates an intense exothermic process with peak centered at ~253°C, while the hydrazine complex decomposes with exothermic process with peaks situated at ~160 and 286°C. Although the thermal decomposition leading to ferrite formation in both the cases is completed at ~300°C, the Cu_{0.9}K_{0.1}Fe₂O₄ obtained from the hydrazine complex showed single spinel phase, Fig. 3. Magnetic characteristic such as saturation magnetization value of 28.5 emu g^{-1} is observed for $Cu_{0.9}K_{0.1}Fe_2O_4$, while the $CuFe_2O_4$ indicated a value of 21 emu g⁻¹. On the other hand a lower than 10 emu/g was observed for the oxides obtained from oxalate precursors, suggesting that a proper cation distribution has taken place in the spinel oxides prepared from the hydrazine method. However, the oxalate precursors of these oxides when decomposed at ~320°C in a controlled atmosphere of a known partial pressure of water vapor (N₂/H₂O/Air) showed higher values of 28 and 38 emu g^{-1} , respectively, for Cu_{0.9}K_{0.1}Fe₂O₄ and $CuFe_2O_4$. Controlled atmosphere of moisture is crucial in stabilizing γ -type phase as observed in FeC₂O₄·2H₂O [1] which easily decomposed to γ -Fe₂O₃ in a known partial pressure of moisture, while in the absence of such atmosphere the product was mainly α -Fe₂O₃.



Fig. 3 X-ray diffraction patterns of copper ferrite and K doped copper ferrite obtained from oxalate hydrazinate precursors

Consideration of this aspect of stabilization of the γ -type of phase was our concern [1–8] and the hydrazine modified precursors found easily stabilize the phase, as the desired partial pressure of moisture was provided by the reaction products. However, in hydrazine complexes the hydrazine liberated at lower temperature reacts with the atmospheric oxygen liberating lot of energy [28],

$$N_2H_4+O_2 \rightarrow N_2+2H_2O; \Delta H = -625 \text{ kJ mol}^{-1}$$
 (1)

The energy thus liberated is sufficient for oxidative decomposition of the dehydrazinated complex and the reaction products, N₂+H₂O, provide the necessary environment of desired partial pressure to stabilize the γ -type phase formation. The, in situ supply of heat by coupled exothermic reaction of hydrazine with oxygen, along with the formation of protective layer of nitrogen and moisture surrounding each particle as a result of reaction product is the key for stabilizing the meta stable γ -type phase. Thus, externally controlled partial pressure of moisture during the decomposition of the oxalate precursors of Cu_{0.9}K_{0.1}Fe₂O₄ and CuFe₂O₄ and in situ supply of the desired partial pressure in hydrazine complexes lead to γ -type phase. The lower saturation magnetization values observed in the ferrite prepared from hydrazine complex required some explanation. The explosive nature of the reaction locally overheats the particle as the protective layer of moisture may not be sufficient for individual particle now acting as micro reactors and the γ -type phase of few particle may transform into α -type.

(MnZn)Fe₂O₄ system

The DSC-TG traces of the oxalate show, Fig.4, an endothermic peak ~218°C that followed by an intense exothermic peak ~275°C, while the hydrazinate complex indicates a broad exotherm in the entire range of temperature, 140-315°C, with a peak positioned at ~185°C. The dehydration followed by decarboxylation is the thermal path for the oxalate, while the dehydrazination and decarboxylation are simultaneous processes in the case of hydrazinate complex. Hydrazine complex yields the spinel phase at



Fig. 4 TG/DSC plots of Mn–Zn Fe-oxalate/oxalate hydrazinate

much lower temperature of about 315°C than that of the oxalate complex which requires more than 350°C, the decomposition products of the oxalate at ~350°C however, show admixture of spinel phase and unreacted Fe₂O₃. Although the spinel formation occurs at lower temperatures in the case of hydrazine method, the saturation magnetization value of 41 emu/g that observed is much lower than the value of 60-80 emu g⁻¹ expected for the ferrite, while the ferrite from oxalate showed mixed phases and saturation magnetization of <10 emu g⁻¹. The lower value may be due to the lack of proper cation distribution and also possibly some Mn^{2+} ions may have oxidized to Mn³⁺ in explosive decomposition of hydrazine precursor. However, a value of 66 emu/g, which is close to 60 emu/g for nanosize $Mn_{0.65}Zn_{0.35}Fe_2O_4$ [29], could be realized for the ferrite synthesized from the oxalate precursor under controlled atmosphere of moisture at ~350°C and then heated at 900°C at a very low partial pressure of oxygen (pO₂ $\approx 10^{-9} - 10^{-15})$ [30].

γ-Fe₂O₃ system

The DSC-TG traces of FeC₂O₄·2H₂O (Fe_oxalate), Fig. 5, show an endothermic peak at ~210°C followed immediately by an intense exothermic peak at ~250°C, while FeC₂O₄·2N₂H₄ (Fe_oxalate_hyd), exothermically decomposes in the entire range of 150–200°C. The decomposition product of the oxalate is hexagonal corundum nonmagnetic α -Fe₂O₃ [1–2] and that of hydrazine complex is the mixture of magnetic cubic spinel, γ -Fe₂O₃, and α -Fe₂O₃. On the other hand, the hydrazine complex, spread on Petri dish, when ignited with a burning splinter it catches fire and a glow forms that then spreads through out the bulk in an ordinary atmosphere, thereby, decomposing autocatalytically mostly into γ -Fe₂O₃ [2].



Fig. 5 TG/DSC plots of Fe-oxalate/oxalate hydrazinate

TiO₂ system

Titanium hydroxide

The TG-DSC traces of titanium hydroxide, TiO(OH)₂·2.5H₂O,(TiOH) and its hydrazinates TiO(OH)₂·1N₂H₄ (UH), [TiOH_hyd(UH)] and TiO(OH)₂·4N₂H₄ (SH), [TiOH_hyd(SH)] are shown in Fig. 6a. All show an endothermic peak ~115°C. The endothermic process in TiO(OH)₂·2.5H₂O is then followed by an exothermic peak ~408°C which suggest that the dehydration is followed by dehydr-



Fig. 6 TG/DSC plots of a – titanium hydroxide/hydrazinate, b – titanium oxalate/oxalate-hydrazinate

oxylation. Though the dehydroxylation occurs at \sim 400°C, the hydroxyl groups still persist in the case of titanium hydroxide prepared by the precipitation of titanium chloride with ammonia, while the hydroxide prepared by the hydrolysis of titanium isopropoxide completes its dehydroxylation below 500°C, Fig. 1. The hydrazinate complex $TiO(OH)_2 \cdot 1N_2H_4$, however, shows an endothermic peak ~115°C followed immediately by an exothermic peak ~150°C and then there occurs another exothermic peak ~378°C. Here the first endothermic is found to be due to dehydrazination as the product that obtained by isothermally heating the sample at ~100°C showed no presence of hydrazine in it. The two consecutive exothermic processes are indicative of exothermic dehydroxylation. The endothermic peak at ~115°C that followed by exothermic peaks ~187 and 401°C in TiO(OH)2. 4N₂H₄ too shows that hydrazine modified hydroxide behaves differently as compared to the hydroxide. The thermal products in all these cases yielded well crystalline anatase, TiO₂, at ~400°C as shown in Fig. 7 by a XRD pattern of anataseTiO₂. Hydrazine method does show lower temperature decomposition and the anatase formed by this method yields yellow color nitrogen doped TiO₂, TiO_{2-x}N_x, [9], while the white pigment grade oxygen deficient nonstoichiometric TiO_{2-x} was the thermal product of the titanium hydroxide. Thus, the hydrazine method not only reduces the temperature of decomposition, but also produces yellow color N-doped visible light sensitive $TiO_{2-x}N_x$ photocatalyst.

Titanium oxalate

The TG-DSC traces of $(NH_4)_2[TiO(C_2O_4)_2] \cdot H_2O$, (Am.Ti.Oxalate), $(N_2H_5)_2[TiO(C_2O_4)_2]$. N_2H_4 (UH) [Hyd Am.Ti.Oxalate(UH)] and $(N_2H_5)_2[TiO(C_2O_4)_2]\cdot 3N_2H_4$ [Hyd Am.Ti. (SH) Oxalate(SH)] are shown in Fig. 6b. (Am.Ti.Oxalate) decomposes giving endothermic peaks at ~114, 198, 272 and 368°C and then exothermic peak at ~500°C. The hydrazine complex prepared under hydrazine [Ti Oxalate hyd (UH)], method, shows an endothermic reaction giving peak at ~114°C and then



undergoes continuous exothermic process that peaks at ~505°C, while the one prepared synthetically, [Ti Oxalate hyd(SH)], indicates endothermic peaks at ~114, 156 and 270°C and then a small exothermic peak at ~372°C that followed by an intense exothermic peak at ~517°C. The thermal product of all these complexes is anatase TiO₂ and the hydrazine complexes yield yellow color N-doped $TiO_{2-x}N_x$ [9]. The XRD pattern of these shows an anatase type phase as in Fig. 7. The nanosize particles are observed in the thermal product of hydrazine complex, whereas, agglomerated particles are found for the decomposed product of oxalate. The TEM of these are shown in Fig. 8. The explosive nature of the low temperature decomposition of the hydrazine complex may be the reason for producing nanosize particles.

Perovskite system

BaTiO₃

The DSC-TG curves of Ba[TiO(C_2O_4)₂]·6H₂O, (BaTi_oxalate) show, Fig. 9, an endothermic peak corresponding to dehydration at ~100°C followed by an exothermic peak at ~487°C. In case of hydrazinate complex, Ba[TiO(C_2O_4)₂]·7N₂H₄, (BaTi_oxalate_hyd) an endothermic peak at ~98°C is followed by a sharp exothermic peak at ~203°C and a broad exotherm in the temperature range 210–400°C. There occurs a small exothermic hump ~450°C. The isother-



Fig. 8 TEM of TiO₂ obtained from a - oxalate hydrazinate, b - oxalate



Fig. 9 TG/DSC plots of BaTi oxalate/oxalate-hydrazinate

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Fig. 10 Dielectric constant of A – Ba Ti oxalate hydrazinate, B – Ba Ti oxalate

mal studies reveal the BaTiO₃ formation at ~450°C in hydrazine complex, while the oxalate complex requires a higher temperature of about 500°C to give the perovskites. The variation of dielectric constant as a function of temperature plots, Fig. 10, of the ferroelectric BaTiO₃ shows an increase in the value with the increase in the temperature up to 115°C and then decreases sharply to zero in the case of the oxide synthesized by hydrazine method. The sample prepared by oxalate route, however, shows a peak $\sim 120^{\circ}$ C, but the increase, as well as the decrease is broad. The Curie temperature thus obtained is as expected for BaTiO₃ and since a sharp increase and decrease in the dielectric constant observed in the oxide prepared by hydrazine route suggests superiority of hydrazine method.

LaCoO₃, LaAlO₃ and GdCoO₃

The oxalate and oxalate hydrazinate precursors of LaAlO₃, La_{0.65}Sr_{0.35}AlO₃, La_{0.8}Sr_{0.2}CoO₃ and La_{0.7}Sr_{0.3}MnO₃ and GdCoO₃ have been synthesized and their thermal decomposition paths have been analyzed [30]. The chemical formulas of few of these precursors are shown in Table 1. The general trend is that the hydrazine complexes show the thermal path different from the oxalate ones, as discussed above. We are not discussing all those here in this paper. However some of the superior behavior of few oxides prepared by the hydrazine method is being described here.

Single phase cubic LaAlO₃, Fig. 11, could be obtained at 1000°C from LaAl(C_2O_4)₃·N₂H₄·3H₂O, while >1400°C is required to achieve the phase for oxalate complex, as well as, for the stoichiometric mixture of La₂O₃ and Al₂O₃. In the figure the highlighted by arrows in the XRD pattern for the sample heated below 900°C are due to La₂O₃ impurity. On the other hand, the Sr doped complex, La_{0.65}Sr_{0.35}Al(C₂O₄)₃·2N₂H₄ heated at 1000°C shows



mixed phases, (not shown here) of the perovskite and However, the single phase La_2O_3 . cubic La_{0.65}Sr_{0.35}AlO₃ could be obtained at ~1500°C. LaAlO₃ is a material used as a substrate for depositing thin films of compatible oxides of superconducting nature and ferroelectric materials and hence the formation of well dense LaAlO3 at ~1000°C from the hydrazine method is quite promising one, as otherwise the ceramic technique using La_2O_3 and Al_2O_3 needs very high temperature, >1400°C [31]. The variation of dielectric constant measured at different temperatures and frequencies, Fig. 12 reveal that dielectric constant of LaAlO₃ has no measurable change in frequency up to 325°C in all frequencies studied from 1 KHz to 1 MHz and the value of 25 observed is close to the expected values of 20–25 [32, 33]. $La_{0.65}Sr_{0.35}AlO_3$, on the other hand, shows the temperature independent dielectric constant of 50 up to 175°C at 1 KHz and 25 till 350°C at 1 MHz, and there appears decrease in dielectric constant with the increase in frequency 1 KHz to 1 MHz. Slightly distorted perovskites $La_{0.8}Sr_{0.2}CoO_{3-\delta}$ with rhombohedral structure could be achieved from oxalate hydrazinate, $La_{0.8}Sr_{0.2}Co(C_2O_4)_3 \cdot 2N_2H_4 \cdot 3H_2O$ ~1000°C, Fig. 13, while the oxalate complex gave no single phase perovskites at this temperature. The electronic conductivity (S cm⁻¹) measured by Van der Pauw method indicated that there occurs insulator-metal (IM) phase change, Fig. 14, around 500 K [34–36] in the $La_{0.8}Sr_{0.2}CoO_{3-\delta}$. However, an oxygen ion conductivity of $1.995 \cdot 10^{-2}$ (ohm cm)⁻¹ observed for La_{0.8}Sr_{0.2}CoO₃ at ~700°C is comparatively much higher as compared to that observed by Wiemhoefer *et al.* [37] of $4.6 \cdot 10^{-5}$ (ohm cm)⁻¹.



Fig. 12 Dielectric constant vs. temperature of $a - LaAlO_3$ and $b - La_{0.65}Sr_{0.35}AlO_3$



Fig. 14 Electronic conductivity of La_{0.8}Sr_{0.2}CoO₃

A gradual IM transition from 500–700 K have been studied [36] by systematic replacement of the R-site in RCoO₃, where R=La/Pr/Nd/Sm/Eu/Gd. These studies were done on the single crystals and since we observed easy formation of the LaCoO₃ and Sr doped LaCoO₃ by hydrazine method, we wanted to extend our preparative method in the above series of the perovskites. And here we are describing the single phase $GdCoO_3$ and Ca/Sr/Ba doped $GdCoO_3$ by hydrazine method.

GdCoO3 system

Chemical formulas of citrate and their hydrazinates have been confirmed by elemental, IR and isothermal total mass loss studies and they are as follows: $GdCo(C_6H_5O_7)_2 \cdot 2H_2O$, $GdCo(C_6H_5O_7)_2 \cdot 4N_2H_4$, $Gd_{0.8}Ca_{0.2}Co(C_6H_5O_7)_2 \cdot 10H_2O$, $Gd_{0.8}Ca_{0.2}Co(C_6H_5O_7)_2 \cdot 4N_2H_4$,

 $Gd_{0.8}Ba_{0.2}Co(C_6H_5O_7)_2 \cdot 10H_2O$,

 $Gd_{0.8}Ba_{0.2}Co(C_6H_5O_7)_2$ ·4N₂H₄. The TG/DSC traces of all these indicate a modification in thermal path with hydrazination, Figs 15a–d. The thermal products of hydrazinate complexes of all these complexes give single phase perovskites at the end of the complete decomposition temperatures, while in the case of citrates further heating is needed to achieve the perovskite after the complete decomposition.

Significance of the hydrazine method of synthesis

The path of decomposition is found to be modified by hydrazinating the hydroxide, oxalate and citrate precursors of the oxide systems studied in the present investigations. In general the hydrazine complexes decompose at lower temperatures than that without the modification. The dehydration is followed by dehydroxylation in the case of hydroxides and the dehydration is followed by decarboxylation in carboxylates, while it is the dehydrazination that is followed by dehydration/dehydroxylation and dehydration/ decarboxylation in the hydrazinated hydroxide and carboxylates, respectively. The dehydrazination that takes place at lower temperatures makes all the differences in the thermal paths of the hydrazinated complexes. It is the hydrazine that is released reacts with the atmospheric oxygen and liberates enormous energy [28], Eq. (1). And this energy is sufficient to oxidative decomposition of the complex devoid of hydrazine, while the complexes without such hydrazine modification have no such an additional energy to decompose and hence they may be decomposing at much lower temperatures. However, it is the reaction products of hydrazine, N₂+H₂O, that provide the required partial pressure of moisture during the decomposition of the $FeC_2O_4 \cdot 2N_2H_4$ [3] to stabilize the γ -Fe₂O₃, as its importance is observed in the thermal decomposition of $FeC_2O_4 \cdot 2H_2O$ [1]. This is due to the fact that FeC₂O₄·2H₂O decomposes in air giving mainly nonmagnetic α -Fe₂O₃,

$$FeC_{2}O_{4}:2H_{2}O \xrightarrow{-H_{2}O} FeC_{2}O_{4} \xrightarrow{CO/CO_{2}} FeO \rightarrow Fe_{2}O_{4} \rightarrow \gamma - Fe_{2}O_{3}) \xrightarrow{\Delta} \alpha - Fe_{2}O_{3}$$
(2)

On the other hand, the decomposition of the FeC_2O_4 ·2H₂O in an inert atmosphere gives Fe_3O_4 , while in a controlled atmosphere of a known partial pressure of moisture the stabilization of γ -Fe₂O₃ takes place



Fig. 15 a, b, c, d TG/DSC plots of Gd (Ca,Ba,Sr)Co-citrate/citrate hydrazinate

$$\operatorname{Fe}_{3}O_{4} \leftarrow \operatorname{FeC}_{2}O_{4} \cdot 2H_{2}O \xrightarrow{N_{2}/H_{2}O/\operatorname{Air}} \gamma \operatorname{Fe}_{2}O_{3}$$
 (3)

And, therefore, $FeC_2O_4 \cdot 2N_2H_4$ during its decomposition easily gets the desired partial pressure of moisture from the reaction products of hydrazine, as well as the thermal products of the oxalate, CO/CO_2 , to stabilize the γ -Fe₂O₃. The energy released by the hydrazine reaction with atmospheric oxygen (Eq. (1)) may be the reason for the easier low temperature decomposition of the hydrazine complexes, but the formation of yellow color nitrogen doped $TiO_{2-x}N_x$ from the hydrazine complexes of hydroxide and oxalate [9] has to be invoked separately. No doubt the hydrazine helps in lowering the decomposition temperature of these complexes, but here the nitrogen that is being formed in hydrazine break up may be getting trapped in the lattice of the titanium oxide. An observation of the N1s peak ~400 eV (Binding energy) in XPS studies of TiO_{2-x}N_x, Fig. 16 and nitrogen content of 0.95% estimated using the oxygen-nitrogen analyzer [16] further supports [9] the introduction of the nitrogen in the TiO₂ by hydrazine method.



Conclusions

- Hydrazine modified hydroxide and carboxylate precursors of the oxide systems: γ-Fe₂O₃, (MnZn)Fe₂O₄, CuFe₂O₄, BaTiO₃, La(Sr)MnO₃, La(Sr)AlO₃, La/Gd(Ca/Ba/Sr)CoO₃, and anatase TiO₂ decompose at low temperatures giving single phase oxide as compared to the ones without such modification.
- Hydrazine released at low temperature during the thermal decomposition reacts with the oxygen in the atmosphere releasing enormous energy,

$$N_2H_4+O_2 \rightarrow N_2+H_2O; \Delta H = -625 \text{ kJ mol}^{-1}$$

that is sufficient for the oxidative decomposition of the hydrazine devoid hydroxide and carboxylate complexes, as compared to the simple hydroxide and carboxylates.

• The reaction products of hydrazine reaction with oxygen not only provide the desired partial pres-

sure of moisture that needed for stabilizing the γ -Fe₂O₃, but also help to incorporate the nitrogen in the lattice of TiO₂ to yield yellow color N-doped TiO_{2-x}N_x photocatalayst.

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